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### *RESEARCH ARTICLE - MATERIALS ENGINEERING*

# **Residual Stresses Characterisation of Hard Ceramic Coating (SiC-5wt%Al2O3) Using X-Ray Diffraction Technique**

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#### **1. Introduction**

High-quality thermal spray coating can be noticed in several applications that incorporate adorable resistance to high temperature, cracking, deboning, and spallation, which reflect primitive coating durability. Thermal spray coating is specifically a useful process that can be used to apply a wide range of materials to a wide variety of substrates. This made a positive contribution in a variety of industries, such as power generation, oil and gas, and medical sectors [1]. However, it should be noted that each industry has its own specific constraints and requirements for thermal spray coatings. For instance, the power generation industry necessitates coatings that can endure high temperatures and corrosion conditions [2]. Also, wear, corrosion, and high temperatures are the main challenges of utilising thermal spray coatings in the oil and gas industry [3]. The medical industry requires coatings that are biocompatible and wear-resistant [4].

Interestingly, residual stresses play a vital role in coating strength. Specifically, residual stresses are internal stresses that occur in a material even in the deficiency of external forces. For instance, tensile residual stresses normally aid to surge the vulnerability to cracking and deboning. Residual stresses can be caused by a diversity of parameters, such as thermal expansion mismatch, phase transformations, and plastic deformation. Residual stresses can meaningfully impact the mechanical properties of a material, such as its strength, toughness, and fatigue life [5, 6].

Multiple colleagues were focused on the assessment of residual stresses in coatings [7-9]. Basically, cooling of a thermal spray coating normally forms tensile or compressive residual stresses. This is belonging to the discrepancy of thermal expansion constants of the substrate and coating [10]. The formed residual stresses are affected by several factors, including the substrate temperature through spray deposition, coating thickness, roughness and porosity [11]. An increase in coating thickness and deposition temperature would positively increase the residual stresses, as experimentally confirmed by Ghafouri-Azar et al. [12]. The most reliable method of thermal barrier coatings (TBC) is the protection of high-temperature gas turbine engines. Such coating is widely demanded due to higher temperature engines being developed [13-15]. This thermal coating is carried out by various mechanisms, such as high-flame temperature methods of plasma spray, flame spray, and arc spray [16, 17]. Solidifying and instantly a few microseconds cooling represent partial and full striking of molten particles onto the substrate's surface. Once the particles are hardened and adhered onto the substrate's surface, the contraction of the splats can be limited. This is carried out by



substrate material or by considered solidifying coating material that causes an inherent deposition or quenching stress tensile stresses. A great theoretical residual stress of up to 1 GPa is caused due to a massively high temperature variance. On the contrary, many relaxation approaches, including the splats' sliding, micro-cracks, plastic distortions, and creeping of material that lead to much lower than 100 MPa of residual stress [18]. In this aspect, X-ray diffraction was noticeably deployed for various types of materials to determine macro and micro residual stresses in a thin film layer. Interestingly, the neutrons' penetration power permits through-thickness stress profiling without the need of any material elimination.

SiC substrate and 95wt.%  $Al_2O_3$  has been applied in several applications as a coating alloy that characterised with low friction and highresistance. Protective coatings against corrosion in steel, aerospace moving constituents, and metal working tools represent some of these industrial applications [19].

Mahmoud et al. [20] used sin2w based on X-ray diffraction (XRD) approach to investigate the residual stresses essentially found in wear protection coatings. Specifically, the coating of wear was conducted via 95% Al<sub>2</sub>O<sub>3</sub> and 5wt.% SiC substrate besides using AlNi alloy as bond coat that being generated via flame spraying method. This in turn has confirmed compressive residual stresses of (-325.67 MPa) for the recent coating. It is noteworthy to mention that (-325.67 MPa) of compression residual stress might be enhanced via the employment of novel mixture of Al2O<sup>3</sup> and SiC substrate.

The development of new thermal spray coatings is a continuing process where several researchers are persistently working to enhance the performance of thermal spray coatings via advancing new materials and processes. Up to the authors' knowledge, the assessment of the residual stresses of SiC substrate and 5wt.% Al2O<sup>3</sup> (this is quite the reverse compositions of the one presented by Mahmoud et al. [20] wear protection coating was not achieved yet. In this regard, the selection of SiC substrate can be attributed to the high hardness and wear resistance of SiC substrate besides its good thermal stability (endures high temperatures without losing its mechanical features), making it ideal for use in protective coatings [21]. Due to its good adhesion property to metal substrates, SiC can be a competitive choice for coatings that are applied to metal components [22]. On top of this, the addition of 5wt.% Al2O<sup>3</sup> would also improve the wear resistance of the coating. Here, it should be noted that most previous studies endorsed the feasibility of using 5wt.% Al2O3. More recently, Sekaran et al. [23] ascertained that the composite containing SiC and 5wt.% Al2O<sub>3</sub> has the most superior mechanical properties of hardness, impact, and tensile strength. Indeed, in the case of hard ceramic coatings, such as SiC substrate and 5wt.% Al<sub>2</sub>O<sub>3</sub>, residual stresses can be predominantly imperative. These coatings are frequently utilised to protect substrates from wear and corrosion. However, if the residual stresses in the coating are too high, they can lead to premature failure of the coating. Therefore, this study intends to resolve this challenge via X-ray diffraction methodology to measure the residual stresses.

The following factors could account for the variance in the residual stresses:

- Differing coefficients of thermal expansion for the substrate, bond coat, and top coat, respectively;
- Breakdown of the coating system;
- The substrate's specific heat capacity;
- The thermal spraying process's kinetic impact of the particle on the substrate, which is brought on by heat transfer from the coating layer to the substrates. All of these factors are vital for the cooling and solidification of the coating layer once the coating process is complete. Surface morphology and coating process factors may also be involved.

#### **2. Materials and Parameters of the Spraying Processes**

Thermal spraying procedure denoted as a flame spraying that used to coatings in air on the plain-carbon steel type AISI 1050. The carbon steel is a cylindrical substrate of 18 mm and 10 mm of diameter and height, respectively. The heat flame is generated via a spray gun in a flame spraying unit as a result of burning of acetylene in the presence of oxygen. Specifically, the melted powder is achieved in the gas mixture and committed to the coated surface using a high-temperature torch of around 3000 °C. To gain and maintain a flame equivalent to the powder's rush speed, it is necessary to manage the pressure of the gases. Therefore, the oxygen pressure must be altered based on the spray gun. This work has been characterised by two top bond coating layers of AlNi alloy. The main intention was to reduce the thermal expansion constant between the substrate and composite ceramic layer. the operating settings are provided in Table 1, while Fig. 1 shows the system of flam spray using the oxygen and acetylene gases mixture to complete the heating and spraying process.

In this regard, it should be noted that the operating parameters of Table 1 were carefully selected based on accumulative experience to maintain the desired thermal coating properties and performance of the composite SiC and 5wt.% Al2O3. For instance, the oxygen pressure of 4 bar was selected to assure melting the SiC and  $\text{Al}_2\text{O}_3$  powder particles while controlling the temperature of the flame. However, any further increase in the oxygen pressure would lead to the oxidation of the powder particles. Furthermore, the acetylene pressure was controlled to be 0.7 bar in a way to control the size and shape of the flame, while providing uniform coverage of the substrate. However, any further increase in the acetylene pressure would cause a carburizing flame. The distance between the flame spraying torch and the substrate was also selected to guarantee a higher heat input without damaging the substrate. This in turn would ensure that the powder particles bond to the substrate.







Fig. 1. The system of flam spray

#### **3. Result and Discussions**

#### *3.1. X-ray diffraction (XRD) phase analysis results*

The X-ray diffraction is a non-destructive method that can be used to measure residual stresses in materials. By measuring the lattice spacing of a material, it is conceivable to decide the amount of strain that exists in the material. This information can then be utilised to measure the residual stresses.

The phase composition of the wear protection coatings depends on the powder, velocity of the particles in-flight, solidification rates, cooling rate and the temperature of the deposited particles. The phase structures of the ceramic coating layers are conducted using X-Ray diffraction technique (XRD), where the X-Ray results show  $\alpha$ Al<sub>2</sub>O<sub>3</sub>,  $\alpha$ -SiC as the projecting phase in pure SiC substrate, respectively, as represented in Fig. 2.



Fig. 2. X-ray diffraction chart for SiC-5wt.% Al2O<sup>3</sup> coating layer

#### *3.2. Scanning electron microscope*

The importance of using a Scanning Electron Microscope (SEM) for various groups is to allocate the surface characteristics. This includes the surface morphology, size and porosity type, cracks and micro-cracks of the coating layers, as depicted in Fig. 3. Specifically, Fig. 3 depicts the SEM morphology of the flame-sprayed SiC-5wt.% Al2O3 coating layer. The coating contains a matrix of SiC substrate and a network of Al2O3 particles. Angular particles of SiC substrate can be seen, while the Al<sub>2</sub>O<sub>3</sub> particles are spherical. The coating contains a number of open and closed pores. The open pores are interconnected and permit for the flow of fluids, in comparison to the isolated closed pores which do not allow for the flow of fluids. The coating also comprises a number of voids, which are empty spaces that are not filled with any material.



Fig. 3. SEM morphology of the flame-sprayed SiC-5wt.% Al2O3 coating layer

#### *3.3. X-ray stress evaluation*

The standard technique of lattice plane distance (d<sub>spacing</sub>) against sin<sup>2</sup>ψ was normally used to determine X-Ray diffraction XRD-based residual stress values [20, 24]. In this regards, sin2 ѱ would denote the residual stresses. XRD is originally investigated using Shimadzu X-Ray Diffractometer type XRD-6000 and CrKα radiation. Moreover, the peak shift of the relevant reflection, which signifies the variation of a lattice plane distance (d<sub>spacing</sub>) of a phase was calculated for 0° and 45° of tilt  $\psi$ -angles. Up to this, a linear regression method of the plot (dspacing) against sin² ѱ used to measure the residual stresses besides estimate the X-ray elastic coefficients. Table 2 lists the physical characteristics of both the coating and the substrate. Furthermore, the coating technique employed in the current study's deposition temperature was 850 °C, the same as modelling for the top coating, bond coat, and substrate. Table 3 presents the gained values from XRD chart.



Table 2. The physical properties of substrate and coating [25-27]

At  $n = 1$ ,  $\lambda = 2.28970$  A<sup>o</sup> and  $\Theta$  (0, 15, 30, 45) degree, the value of d can be calculated using the Brag Law (2d sin $\theta = n\lambda$ ). The linear slope of the plot d<sub>spacing</sub> against sin<sup>2</sup>  $\psi$  can be estimated from Fig. 4.

Eq. 1 used to calculate the stress

$$
\sigma = \frac{E}{(1+v)} \frac{1}{d\sigma} \frac{\partial d}{\partial \sin^2 \psi} \tag{1}
$$

 $\sigma$  is the elastic module, V is the poussin ratio, d is the dspacing. Fig. 3 shows the slop  $\partial d/(\partial \sin 2\psi)$  = -0.00876. Therefore, dspacing values from XRD and CrKα radiation are extracted from Eq. 1. Occasionally, Eq. 1 identifies that SiC-5wt.% Al2O3 topcoat was exposed to maximum compressive residual stresses of (-594.029 MPa). In this regard, Mahmoud et al. [20] concluded that the compressive residual stresses of the contrast composite of 5wt.% SiC substrate and 95%  $Al_2O_3$  is (-325.67 MPa). This difference of the compressive residual stresses between these two composites can be attributed to the difference in the thermal expansion coefficient between the associated compounds of the coating and substrate (SiC and Al2O<sub>3</sub>). SiC and Al2O<sub>3</sub> have thermal expansion coefficients of  $4.0x10^{-6}/K$ , and  $8.5x10^{-6}/K$ , respectively. This means that SiC expands less than Al<sub>2</sub>O<sub>3</sub> when heated. Referring to the composite used in this research of 95wt.% SiC substrate and 5% Al<sub>2</sub>O<sub>3</sub>, it is fair to expect that the higher volume fraction of SiC in the coating can generate the higher difference in thermal expansion between the SiC and Al2O<sup>3</sup> particles. In turn, this would lead to a greater degree of compression and, therefore a higher compressive residual stresses compared to the low fraction of SiC in the composite of 5wt.% SiC substrate and 95% Al<sub>2</sub>O<sub>3</sub> of Mahmoud et al. [20].

These residual stresses were assessed in the surface layer of coatings adherent to the substrates. Furthermore, 500 µm thickness of layers was noticed that compatible to the X-ray penetration. More importantly, several sites of each coating were tested in the current study besides determining the residual stresses along two perpendicular directions matching to azimuth angles of 0 and 90º. For each sample examined, the results identified the characteristics of a plane-equi-axial and compressive stress state, with fixed values at locations far from borders or irregularities. The upcoming results of this study would confirm these characteristics.

The top surface residual stresses can be determined using the SiC-5wt.% Al2O3 coating standard that is applied via flame spraying. The three most important factors in this regard are the substrate's composition (AISI 1050), thickness (10 mm), and bond coat (200 m) of NiAl. As a result of widespread micro-cracking caused by spraying, the level of residual strains remained constant. The stress release is significantly impacted by the coating imperfections, porosities, and micro-cracks, and only quenching stresses are left in the final deposit [18].

Interestingly, the above research presents the deployment of X-ray diffraction method to identify the residual stresses in ceramic coating of SiC-5wt.% Al<sub>2</sub>O<sub>3</sub> in an acceptable accuracy. To ascertain the real loads that a component experiences, it is crucial to understand the residual stress conditions. Changes in residual stresses can directly influence a diversity of mechanical properties of a material, including its strength, toughness, fatigue life, corrosion resistance and wear resistance. For instance, high residual stresses can cause a reduction in strength and toughness besides increasing fatigue life and corrosion resistance. However, compressive residual stress on a component's surface is often advantageous. It usually increases fatigue strength and fatigue life, slows the spread of cracks, and increases resistance to environmental cracking, such as hydrogen-induced cracking and stress corrosion cracking. Tensile residual stress on the component's surface is often not desired since it reduces fatigue life and fatigue strength, speeds up crack growth, and weakens resistance to environmentally induced cracking.

Referring to the above discussion, it should be noted that any changes in residual stresses can influence the potential applications of a composite material. For instance, a composite material with high residual stresses is not a practical option for applications where strength and toughness are important. However, this composite material can be useful for applications where fatigue life and corrosion resistance are important. Accordingly, the compressive residual stresses that were measured in this study for SiC-5wt.% Al2O<sup>3</sup> hard ceramic coating, are advantageous for many applications to enhance the strength, toughness, and wear resistance of the coating.



Fig. 4. Relationship between  $\sin^{2} \psi$  and dspacing

#### **4. Conclusion**

The X-ray diffraction method is proven to be a feasible tool to evaluate and analyse the residual stresses distributions in composite wear protection coating units. In this study, the stress assessment of complicated coating systems necessitates an advanced approaches of coating and the substrate compared to coating of a single film on substrate systems. Specifically, the recent study has carried out the stress analysis of a single layer of SiC-5wt.% Al<sub>2</sub>O<sub>3</sub> flame spraying composite coating. This in turn affirmed the maximum value of compressive residual stresses of (-594.029 MPa), which was determined using the least square-fitting method that characterised by the related function to the experimental sin2ψ-data. However, irregular distribution of the residual stresses throughout the thickness was noticed within the composite ceramic top coating. Specifically, the residual stresses have attained its maximum value at the interface and its minimum value out the interface on the free surface.

The findings of this study can provide a better perception of the residual stresses that exist in hard ceramic coatings. This would therefore enable to design coatings that are more resistant to failure. Also, the findings can aid to develop new techniques to measure the residual stresses in hard ceramic coatings while improving the quality control of these coatings. On top of these advantages, the current findings can be utilised to improve new applications of hard ceramic coatings to specifically protect components in harsh environments in a wide set of industries such as aerospace and automotive industries. However, the investigation of the optimum concentration of Al2O3 is feasible to be investigated in the upcoming research.

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