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RESEARCH ARTICLE – NANOTECHNOLOGY

Calcium Phosphate Coatings with Controlled Micro/Nano-Structures for Endothelial Cells Viability

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Article Info.	Abstract
Article history:	Hydroxyapatite (HA) scaffolds produced by the accumulation of HA fibers were separately treated hydrothermally in three calcium phosphate solutions to form coatings of different micro/nano-structures. Different micro/nano-structure and
Received 30 March 2024	morphologies have been regulated on the surface of treated HA scaffolds. Plate-like compromise flower-like morphology was obtained with solution 1 (Ca-sufficient) i.e., ratio: Ca/Ca=1%; Ca/P=1.67. Full coatings (flower-like) morphology treated after Cu-doped coating solution 2 (Cu/(Cu+Ca) = 5%; ratio: (Cu+Ca)/P = 1.67). Furthermore, partial coatings
Accepted 22 May 2024	(flower-like) morphology fabricated with solution 3 (Ca-deficient and Cu-replacement), i.e., ratio: Ca/Ca=0.95%; Ca/P=1.58. The results showed the effect of hydrothermal coatings on HA scaffolds. Cultured human endothelial cells spread and proliferated better on the treated HA scaffolds than on the uncoated scaffolds, suggesting a potential effect of
Publishing 30 June 2024	calcium phosphate surface morphology on endothelial cell response. Thus, it can provide an appropriate micro/nano- structure approach supporting angiogenesis capacity, which is a necessity to accelerate the time of bone healing and regeneration.

Keywords: Hydroxyapatite; Micro/Nano-Structure; Scaffolds; Coating; Endothelial Cell.

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1. Introduction

Hydroxyapatite (HA, $Ca_{10}(PO_4)_6(OH)_2$) is the common mineral found in vertebral bones [1], and widely used for bone repair because of its similarity to human bone mineral [2, 3]. In addition to the chemical constitution, an ideal surface micro/nano-structure is also desirable for the development of bioactive materials [4]. In particular, Ca cations in the HA structure can be substituted by several metal cations, e.g., Na⁺, Ag⁺, Mg²⁺, Ni²⁺, Sr²⁺, Fe³⁺ and La³⁺[5]. Various techniques have been reported for surface coating or modification of calcium phosphate scaffolds, such as precipitation, sol-gel, solid-state, hydrothermal, and biomimetic methods [6-8].

Of these, the hydrothermal process offers advantages such as fast production, relative technical simplicity, and high crystallinity of the product [7]. In this method, the pH and ion concentrations (*e.g.*, Ca^{2+} , PO_4^{3-}) are key parameters affecting the morphology, and crystal size of the coating formed [9]. However, few studies have investigated modulating the micro/nano-structure of calcium phosphate coatings deposited on HA scaffolds. Most of HA scaffolds modifications in hydrothermal method was used inorganic [10], or organic [11] modifiers. Our previous study reported controlling the nanostructures of calcium phosphate coatings hydrothermally deposited on HA scaffolds by introducing organic crystal growth modifiers (*i.e.*, hexacarboxylic acid) [6]. As many organic modifiers (e.g., hexacarboxylic acid) are generally not present in the human body, their use might involve concerns of safety and clinical practicality. Recently, also we prepared micro/nano-structure on HA scaffolds by hydrothermal treatment with the assistance of organic modifier ions (copper)[10]. This work investigated the effect of Cu^{2+} on the HA scaffolds coating by the used hydrothermal method. But, did not consider the impact of the identical hydrothermal condition without an inorganic modifier (copper ions) on HA scaffolds.

Addressing the micro/nano-structure of HA scaffolds with hydrothermal coatings and their effects to endothelial cell (EC) viability is still unclear. In contrast, methods of modifying the surface morphology of HA scaffolds with only physiologically present substances may be preferred. Furthermore, angiogenesis formation has an essential role in physiological bone defect repair [12]. Consequently, the interactions between endothelial cells (EC) and orthopedic biomaterials need to be investigated for achieving optimum clinical outcomes. Several studies

Nomenclature & Symbols					
HA	Hydroxyapatite	EC	Endothelial Cell		
Ca(NO ₃) ₂	Calcium Nitrate	Na ₂ HPO ₄	Disodium Phosphate		

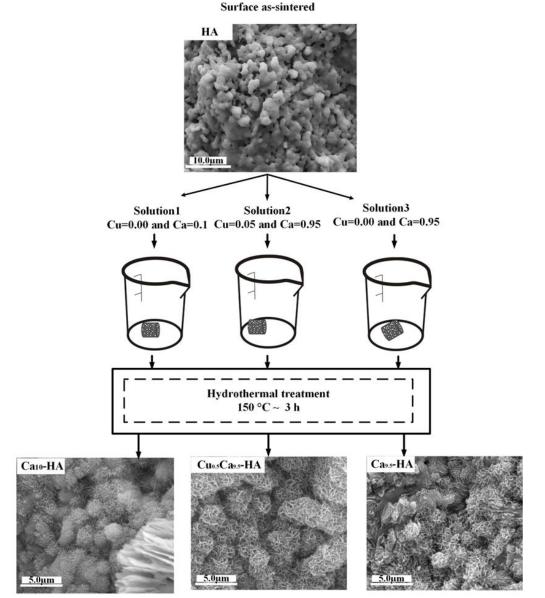
examined EC interactions with HA granules or nanoparticles, and reported favorable responses such as normal morphology and activated gene expression related to angiogenesis [13, 14].

In this study, a hydrothermal treatment was used to prepare calcium phosphate coatings with controlled micro/nano-structures on HA scaffolds. The physiological effects of the coatings with and without copper ions have been studied. Furthermore, the adhesion of EC on the surfaces was studied in vitro, and their formation mechanisms were examined.

2. Materials and Methods

2.1. Material HA scaffolds preparation

An HA slurry was prepared by mixing 20 g of HA *powder* (Kelong Chemical, Chengdu, Sichuan, China) with 100 mL of sodium alginate solution (3 wt.%) [15]. The slurry was transferred in a syringe and injected into a CaCl₂ solution (0.2 mol/L) to form cross-linked fibers. The fibers were immediately collected and packed in molds to create cylindrical scaffolds. They were dried (70 °C, 10 h) and sintered eliminating at (1200 °C, 2 h) [15] to form scaffolds (size: Φ 5×10 mm). Three calcium phosphate coatings, termed Ca_{9.5}-HA, Ca₁₀-HA, and Cu_{0.5}Ca_{9.5}-HA were separately prepared as shown in Fig. 1.



Mico/nano-structed surfaces of treated HA scaffolds

Fig. 1. Schematic illustration of the hydrothermal coatings process and the resulting scaffold surface morphologies

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Three solutions were prepared for coating preparation as showed in Table 1. The first solution (Solution 1), comprising 0.1 mol/L Ca(NO₃)₂ and 0.06 mol/L Na₂HPO₄ (Ca/P molar ratio=1.67), was used for the preparation the coatings on the Ca₁₀-HA scaffolds. The second one (Solution 2), containing 0.095 mol/L Ca(NO₃)₂, 0.005 mol/L Cu(NO₃)₂, and 0.06 mol/L Na₂HPO₄, (Cu+Ca)/P=1.67; and Cu/(Cu+Ca)=5%) was used to prepare the coatings on Cu_{0.5}Ca_{9.5}-HA scaffolds. The third one (Solution 3), containing 0.095 mol/L Ca(NO₃)₂ and 0.06 mol/L Na₂HPO₄, (Ca/P=1.58) was used to prepare the coatings on Ca_{9.5}-HA scaffolds. Each solution was adjusted to pH 2.50 ± 0.05 by dropwise addition of HNO₃; then, 1 g/30mL of urea (CH₄N₂O) was added to the solution [16]. Then, the as-sintered HA scaffolds were immersed separately in the three solutions and transferred in an autoclave. The autoclave was heated to 150 °C, maintained at this temperature for 3 h, and then cooled to room temperature, as schematized in Fig. 1. During this hydrothermal treatment, urea decomposed to generate ammonia, thereby increasing the pH and inducing calcium phosphate crystallization. Finally, the samples were retrieved, rinsed with deionized water, and dried (80 °C, 3 h) [6].

Samples	Ca	Cu	Р	Ca/P	(Cu+Ca)/P	Cu/(Cu+Ca
Ca ₁₀ -HA	39.86	0	18.4	1.67	1.67	0.00
Cu0.05Ca9.5-HA	37.87	3.15	18.4	1.67	1.67	0.05
Ca9.5-HA	37.87	0	18.4	1.67	1.58	0.00

2.2. In vitro cell culture preparation

Human ECs (Sichuan University, Chengdu, Sichuan, China) were seeded on scaffolds (pre-sterilized at 140°C, 3 h) at 1×10^5 cells/scaffold, and cultured (37 °C, 5% CO₂, 95% relative humidity) in 80% of α -minimum essential medium (MEM; Gibco, Carlsbad, CA, USA) supplemented with 10% fetal bovine serum, and 1% penicillin/streptomycin for 5 d. Alamar blue (AB) assays were performed to determine the proliferation of ECs. The medium was discarded and replaced with fresh medium with 10% AB reagent (Invitrogen, Carlsbad, CA, USA). After incubation at 37 °C for 4 h, the optical densities at 570 and 600 nm were measured (µQuant microplate reader, Bio-Tek Instruments, Winooski, USA). Wells with as-sintered samples were used as the control, and empty wells served as the blank. Data were expressed as a percentage of AB reduction, which correlates with the number of viable cells.

3. Result and Discussion

3.1. Microstructure formation on HA scaffold

The surface microstructure of the un-coated and coated scaffolds was investigated by means of Scanning electron Microscope (SEM). Fig. 2 are showed un-coated scaffolds, which are covered with smooth crystalline grains, included micro/nano-crystals and pores with an average crystal's diameter is: 2.1 ± 1.0 µm.

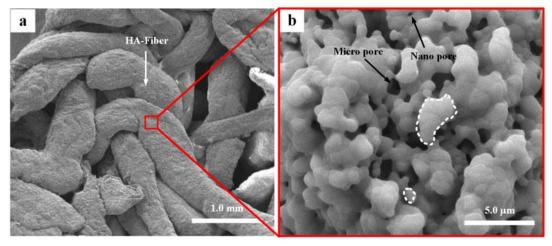


Fig. 2. SEM micrographs of (a, b) typical morphologies of as-sintered HA

After hydrothermal treatment by used Ca/P solutions (Fig. 3), the morphology of treated scaffolds was substantially changed compared with the un-coated one. Figs. 3a and b illustrated plate-like micro-crystals on the surface of Ca₁₀-HA scaffold (crystals diameter: $46.45 \pm 20 \mu$ m), aggregating into flower-like micro/nano-structures (crystal's diameter is: $2.21 \pm 0.5 \mu$ m). The results indicated that the Ca-sufficient in solution 1 could affect the size of crystals formations on Ca₁₀-HA scaffold compared with others. Fig. 3 demonstrated the improvmet of coatings on the Cu_{0.5}Ca_{9.5}-HA scaffold surfaces. Accommodated scattered of flower-like (crystals diameter size approximately $3.4 \pm 1.4 \mu$ m), and silkworm-like (crystal's diameter is: $8.5 \pm 2.1 \mu$ m) as shown in Fig. 3c. The result indicted the effect of Cu on the hydrothermal solution (Cu-presence), which it could facilitate full coatings on the Cu_{0.5}Ca_{9.5}-HA scaffold surfaces (Fig. 3c). While, Fig. 3d showed Ca_{9.5}-HA scaffold with partially coatings compared with Cu_{0.5}Ca_{9.5}-HA scaffolds. The Coating by used Cu-replace solutions could change the scaffolds morphology from smooth like (uncoated scaffold) to the flower-like (crystal's diameter is: $2.4 \pm 0.8 \mu$ m) and silkworm-like (crystal's size is: $4.03 \pm 1.3 \mu$ m). Moreover, the black rows in Fig.3d demonstrated the partial coating area on the scaffold.

In general, the growth of calcium phosphate crystals is depend to the solution pH, the temperature; the Ca/P ratio and the presence of ions or biomolecular [17].

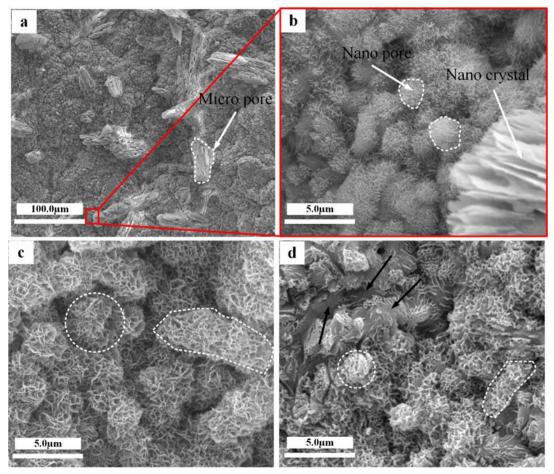


Fig. 3. SEM characterized the coated scaffolds hydrothermally, micrographs of (a, b) Ca10-HA, (c,) Ca9.5-HA, and (d) Cu0.05Ca9.5-HA

Based on the above results, we predicted the mechanism of coatings behind calcium phosphate deposited on the treated HA scaffolds. When, as-sintered HA scaffolds were immersed into solutions, under the initial acidic conditions (pH = 2.5), Ca^{2+} and $PO4^{3-}$ probably dissolved into the surrounding solution from the scaffolds. With an increase in the reaction temperature, the solution became increasingly alkaline because of urea decomposition. The dissolved ions may accelerate crystal nucleation around the surface of the scaffolds; thus, process is promoting the Ca/P micro/nano-structure deposited on the surface of the HA scaffolds. A similar experiment was carried out by Ye et al. [8], he used hydrothermal conditions to generate HA micro-whiskers in Ca/P ceramics. The whisker growth mechanism obtainable a dissolution-redeposition procedure, which specifically indicated that β -TCP in the ceramic matrix simply dissolved into calcium and phosphate ions, then redeposited and shaped HA micro-whiskers under hydrothermal treatment.

The overall results suggested that the micro/nano-structures deposited on the treated scaffolds are attributed to the effect of solutions compositions. The surfaces of the scaffolds may trigger the EC viability response and may enhance the angiogenesis capacity as well as bone healing and regeneration.

To further characterize the surface composition of the treated HA scaffolds, used X-ray diffraction (XRD, PANalytical X'Pert PRO, Philips, Netherlands, CuK α , 35 mA, 45 kV). The XRD patterns (Fig. 4) illustrated the treated scaffolds which had a compose of apatite phase similar to the phase that constituted an initial HA scaffold. Furthermore, no significant difference was observed for Cu_{0.5}Ca_{9.5}-HA respect to the addition of copper ions, which may be attributed to the strong diffraction peak of the substrate and lower presence of copper.

3.2. In vitro evaluation of HA and treated scaffolds

The EC morphologies on treated HA scaffolds was investigated by SEM. Fig.5 dementated a sustainable increase of EC proliferation on HA scaffolds exhibited with increasing of culture time. Fig.5a illustrated the cells had distributed in area: 0.17 μ m on the HA scaffold (uncoated). Fig. 5b showed the EC were spread in area: 0.48 μ m on Ca₁₀-HA scaffold. Furthermore, Fig.5c illustrated the spread of EC on Ca_{9.5}-HA scaffold, with more adhering cells cover an area: 0.35 μ m. Fig.5d showed EC were dispersed in area: 0.87 μ m on Cu_{0.5}Ca_{9.5}-HA scaffold.

The results indicated that the $Cu_{0.5}Ca_{9.5}$ -HA scaffold has a higher presence of EC spread compared with the others, and the less EC spread was observed on HA scaffold (un coated). These findings indicated that the treated scaffolds can support EC adhesion and spread better than the un coated scaffold, which may attribute to their structures [18, 19]. Furthermore, the Cu-presences could not only influence the physical-chemical properties of scaffolds, but also it can support EC adhesion and spread on it [20].

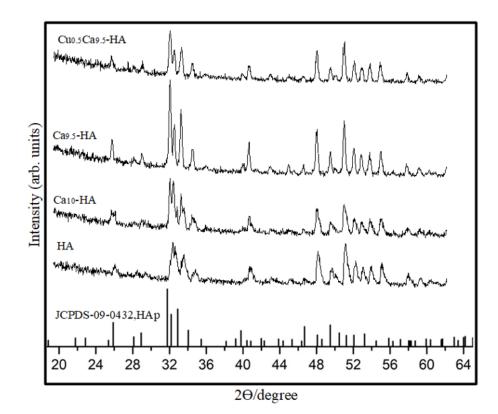


Fig. 4. XRD spectra of as-sintered HA, Ca₁₀-HA, Ca_{9.5}-HA, Cu_{0.5}Ca_{9.5}-HA, and Standard spectra of (HAp)

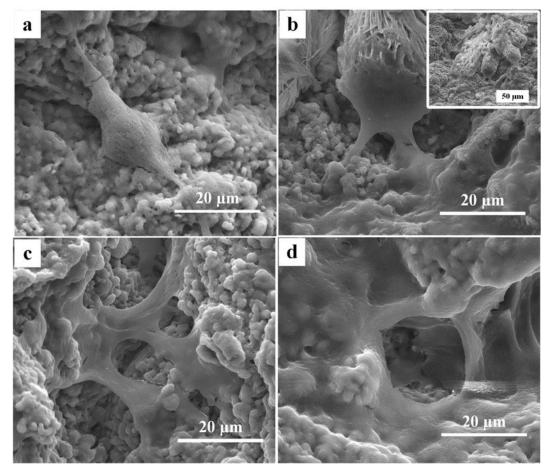


Fig. 5. SEM images illustrated endothelial cell (EC) culture after five days: (a) as-sintered HA scaffold;(b) Ca₁₀-HA scaffold; (c) Ca_{9.5}-HA scaffold; and (d) Cu_{0.5}Ca_{9.5}-HA scaffold

3.3. EC Viability of HA and treated scaffolds

Fig. 6 illustrated Alamar Blue (AB) assays characterized on day 1, 3, and 5. The difference between groups occurred on the 3^{rd} and 5^{th} days. The t-test value showed a significant difference between groups (p < 0.05). Within the same group, a different observation was illustrated in the EC viability rate during culture times. The uncoated HA scaffolds was used as control in this experiment. After culturing the EC viability for 1st day, the scaffolds Ca₁₀-HA and Ca₉.5-HA exhibited insignificant differences compared with control (p <0.05). The result found that the Ca₁₀-HA (Ca- sufficient) has affected the EC viability response compared with Ca_{9.5}-HA (Ca- deficient), thus may imply that the key factor of this effect is the micro/nano-structures growth on Ca₁₀-HA scaffold. The EC viability response on Cu_{0.5}Ca_{9.5}-HA scaffold after 5th day was significantly different compared with the other groups (p < 0.05). The results indicated that Cu_{0.5}Ca_{9.5}-HA scaffold was obtained a higher EC viability compare with the others scaffolds. Thus, refering to the desire micro-nano-structure on the scaffold, which has prepared with Cu-doped hydrothermal coating solution. Previous study reported that the copper ions as a trace element in the human body can play a vital role in promoting endothelial cells migration and the angiogenic formation [21]. The overall results suggested that the Cu_{0.5}Ca_{9.5}-HA scaffolds especially in uses of bone healing and regeneration.

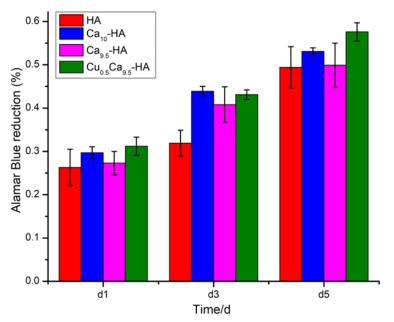


Fig. 6. The bar chart illustrated EC viability during cell culture

4. Conclusion

- Calcium phosphate coatings with controlled micro/nano-structures were successfully growth on HA scaffolds by used hydrothermal method.
- The results indicated that solution1 (Ca-sufficient) was improve different morphologies (micro/nano-structure) on Ca10-HA scaffold, compared with the morphology improved on Ca9.5-HA scaffold, thus treated with solution 2 (Ca- deficient).
- The results indicated that Cu-doped hydrothermal coatings could improve the structure of Cu0.5Ca9.5-HA scaffold and influence the full coatings on scaffold compared with Ca9.5-HA scaffold (without copper).
- The result found the treated Cu0.5Ca9.5-HA and Ca10-HA scaffolds are more convenient for EC viability compared with the other scaffolds.
- The over all results, suggeted that the micro/nano-structure on the scaffolds may provide an appropriate approach for angiogenesis capacity.

Conflicts of Interest

The authors declare no conflicts of interest.

Acknowledgments

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